

## Preparation of Head-to-tail $\pi$ -Conjugated Poly(thiophene-pyridine) and Polypyrimidine by Organometallic Polycondensation

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The metalation of Br-ThPy-Br (Th: thiophene-2,5-diyl; Py: pyridine-2,5-diyl) and 2-iodo-5-bromopyrimidine with *i*-PrMgCl proceeds regioselectively. The cross-coupling polycondensation of regioselectively metalated intermediates gives head-to-tail  $\pi$ -conjugated poly(thiophene-pyridine) and polypyrimidine, respectively, and their packing structures in the solid state and optical data are discussed.

The synthesis of regioregular  $\pi$ -conjugated heteroaromatic polymers by cross-coupling polymerization is important for controlling the solid structures and electrical and optical properties of  $\pi$ -conjugated polymers.<sup>1,2</sup> Head-to-tail-type poly(3-alkylthiophene-2,5-diyl) (HT-P3RTh)<sup>1,2</sup> and poly(6-alkylpyridine-2,5-diyl) (HT-P6RPy)<sup>3</sup> (Chart 1) have been prepared via the regioselective metalation<sup>1-4</sup> of dihalo starting materials (e.g., via XMg-C<sub>4</sub>HS(R)-X, X = halogen) and cross-coupling polymerization using Ni(II) catalysts.

Here, we report on the selective metalation of the following dihalo compounds with *i*-PrMgCl and the utilization of the metalated intermediates for the synthesis of new head-to-tail-type  $\pi$ -conjugated polymers.

Reactions of **1** and **2** with *i*-PrMgCl<sup>4</sup> gave the intermediates **3** and **4** (Scheme 1). The hydrolysis of the formed intermediates gave only H-Th-Py-Br and 5-bromopyrimidine, respectively,

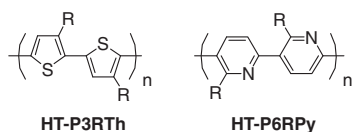
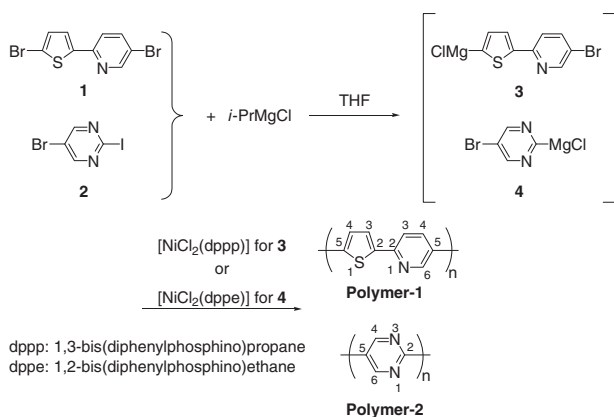


Chart 1.



Scheme 1.

and no Br-Th-Py-H or 2-iodopyrimidine was detected in the hydrolysis products by <sup>1</sup>H NMR spectroscopy. The selective formation of **3** was also supported by the production of the following boron compound **5** (Chart 2) from **3** in good yield.<sup>5</sup> These results support the selective metalation of **1** and **2** by *i*-PrMgCl. The use of 2,5-dibromopyrimidine, instead of **2**, did not lead to the selective formation of **4**.

The addition of diphosphine complexes of nickel to **3** and **4** in THF gave **Polymer-1** and **Polymer-2**, respectively.<sup>5</sup> On the other hand, the zerovalent-nickel-complex ([Ni(0)L<sub>m</sub>]) promoted dehalogenative polycondensation of dihaloaromatic compounds usually gives head-to-head-rich regiorandom polymers,<sup>6</sup> and such polymers, i.e., **Polymer-1'**<sup>7</sup> and **Polymer-2'**,<sup>8</sup> were previously prepared by the dehalogenative polycondensation of **1** and 2,5-dibromopyrimidine, respectively.

**Polymer-1** was soluble in HCOOH and showed an intrinsic viscosity [ $\eta$ ] of 0.58 dL g<sup>-1</sup> (dL = 100 mL) in HCOOH. The Suzuki cross-coupling polycondensation of **5** using [Pd(PPh<sub>3</sub>)<sub>4</sub>] and K<sub>2</sub>CO<sub>3</sub> in THF gave **Polymer-1** with a smaller [ $\eta$ ] of 0.16 dL g<sup>-1</sup>. **Polymer-2** was soluble in H<sub>2</sub>SO<sub>4</sub> and partly soluble in HCOOH; however, it was insoluble in the other solvents tested. The MALDI-TOF-Mass spectrum of **Polymer-2** showed peaks in the range from *m/e* = 450 corresponding to *n* = 6 in (C<sub>4</sub>H<sub>2</sub>N<sub>2</sub>)<sub>n</sub> to *m/e* = 1200 (*n* = 15).

The <sup>1</sup>H NMR spectra of **Polymer-1** and **Polymer-1'** (cf., Figure S1 in Supporting Information<sup>5</sup>) clearly differ from each other owing to the difference in microstructure between the two polymers. Because **Polymer-1** has only aromatic protons, detailed analysis of its <sup>1</sup>H NMR spectrum was difficult. However, the following hexyl polymer (**Polymer-3a** (Scheme 2)<sup>5</sup>) synthesized analogously gives rise to a simple <sup>1</sup>H NMR spectrum (Figure 1), supporting the formation of a regioregular polymer. In particular, the appearance of the  $\alpha$ -CH<sub>2</sub> signal, which is often used for evaluating the regioregularity of  $\pi$ -

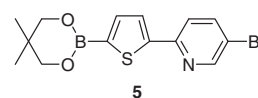
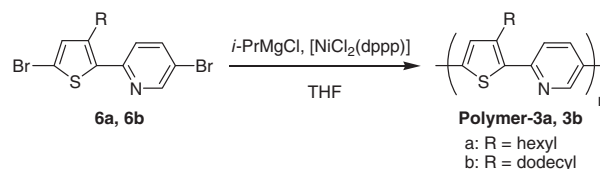


Chart 2.



Scheme 2.

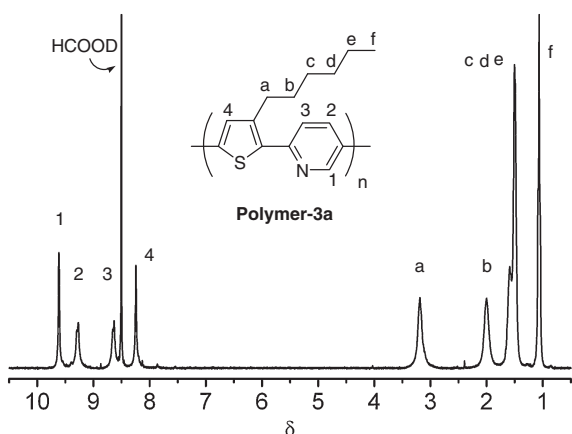


Figure 1.  $^1\text{H}$ NMR spectrum of **Polymer-3a** in  $\text{DCOOD}$ .

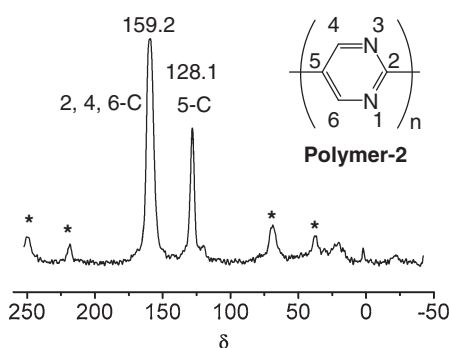


Figure 2. CP/MAS solid  $^{13}\text{C}$ NMR spectrum of **Polymer-2**. Peaks with \* are spinning side bands.

conjugated polymers with an alkyl side chain,<sup>1–3,9</sup> as a single peak supports the regioregular structure of **Polymer-3a**; regio-irregular polymers usually show split  $\alpha\text{-CH}_2$  signals. The  $^1\text{H}$ NMR spectrum of **Polymer-3a** is compared with that of a regioregular alternating  $\pi$ -conjugated copolymer of alkylthiazole (Tz(R)) and thiophene.<sup>9</sup> **Polymer-3b** also gives rise to a simple  $^1\text{H}$ NMR spectrum, which indicates its regioregular structure.

**Polymer-3a** and **Polymer-3b** are soluble in *o*-dichlorobenzene and 1,1,2,2-tetrachloroethane. **Polymer-3a** and **Polymer-3b** showed  $M_w$  (weight-average molecular weight) of 11000 and 56000 with polydispersity indexes of 3.6 and 3.1, respectively, in GPC analysis at 140 °C (vs. polystyrene standards, eluent: 1,2,4-trichlorobenzene).

Figure 2 shows the CP/MAS solid  $^{13}\text{C}$ NMR spectrum of **Polymer-2**. A sharp signal at  $\delta$  128.1 is assigned to the 5-C carbon of the pyrimidine ring. 2-C, 4-C, and 6-C carbons show a single peak at  $\delta$  159.2. The  $^{13}\text{C}$ NMR spectrum pattern of **Polymer-2** is similar to that of **Polymer-2'**.<sup>8</sup> However, the 5-C peak of **Polymer-2'** is broader (half width = 376 Hz) than that of **Polymer-2** (half width = 312 Hz), owing to the regio-irregular structure of **Polymer-2'**. On the other hand, for the head-to-head (HH)-type poly(pyrimidine-2,5-diyl) (**HH-PPyrim**) separately prepared from 2,2'-dichloro-5,5'-bipyrimidine,<sup>10</sup> the two  $^{13}\text{C}$ NMR peaks are shifted upfield (155 and 126 ppm, respectively).

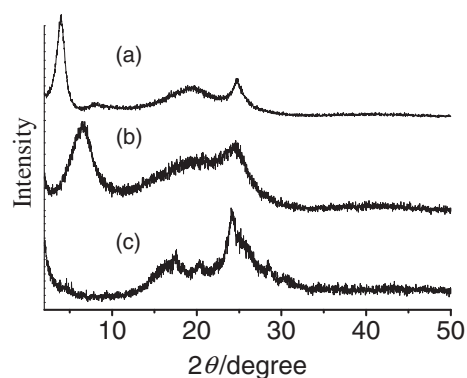
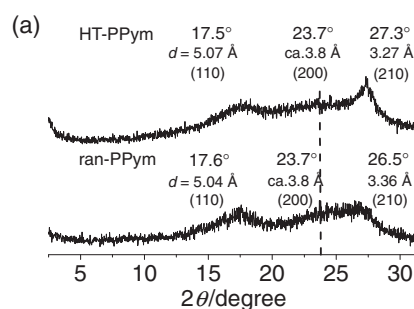


Figure 3. Powder XRD patterns of (a) **Polymer-3b**, (b) **Polymer-3a**, and (c) **Polymer-1**.



(b)

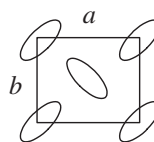
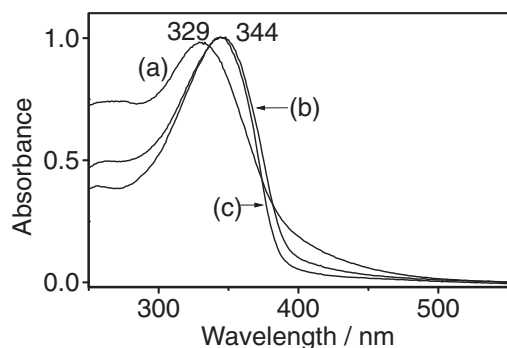


Figure 4. (a) Powder XRD patterns of **Polymer-2** and **Polymer-2'**. (b) Postulated herringbone packing of **Polymer-2**.<sup>11</sup>

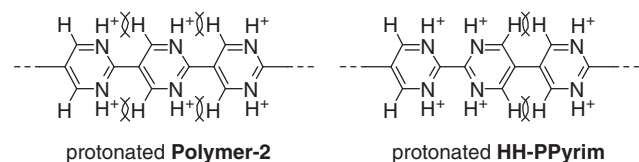
Figure 3 shows the powder X-ray diffraction (XRD) patterns of **Polymer-1**, **Polymer-3a**, and **Polymer-3b**. As shown in Figure 3, the XRD patterns show distinct peaks supporting the notion that the polymers assume ordered structures in the solid state. The XRD pattern of **Polymer-1** is considerably different from that of **Polymer-1'** shown in Figure S2.<sup>5</sup>

The XRD patterns of **Polymer-3a** and **Polymer-3b** resemble those of regioregular  $\pi$ -conjugated aromatic polymers with long alkyl side chains,<sup>1–3</sup> including the regioregular alternating copolymer of Tz(R) and Th (see above).<sup>9</sup> The  $d$  space of the peak in a low-angle region is usually assigned to the distance between polymer chains separated by alkyl side chains. By increasing the length of an alkyl side chain from hexyl ( $2\theta = 6.3^\circ$ ,  $d = 14 \text{ \AA}$ ) to decyl ( $2\theta = 3.8^\circ$ ,  $d = 23 \text{ \AA}$ ), the  $d$  space increases reasonably.

The comparison of the XRD patterns of **Polymer-2** and **Polymer-2'** shows differences, as shown in Figure 4. The XRD pattern of **Polymer-2** shows sharper XRD peaks than that of **Polymer-2'**. Linear  $\pi$ -conjugated polymers without side chains, e.g., poly(*p*-phenylene) and poly(thiophene-2,5-diyl), often



**Figure 5.** UV-vis spectra of (a) **Polymer-2**, (b) **HH-PPyrim**, and (c) **Polymer-2'** in concd  $\text{H}_2\text{SO}_4$ .



**Figure 6.** Postulated protonation of **Polymer-2** and **HH-PPyrim**.

assume herringbone packing in the solid state,<sup>1</sup> and the packing of **Polymer-2** may also be explained by the herringbone packing in view of its XRD pattern and density, as shown in Figure 4.<sup>11</sup>

Both **Polymer-1** and **Polymer-1'** showed a UV-vis peak at about 475 nm in  $\text{HCOOH}$ . **Polymer-1** obtained from **5** and with a smaller  $[\eta]$  gave rise to a UV-vis peak at 455 nm in  $\text{HCOOH}$ . **Polymer-3a** showed a UV-vis peak at 438 nm in  $\text{HCOOH}$ . Figure 5 shows the UV-vis spectra of **Polymer-2** and related polymers in concd  $\text{H}_2\text{SO}_4$ . The  $\pi$ - $\pi^*$  transition peaks of polypyrimidines appear at approximately 335 nm, which is considerably shorter than that (ca. 370 nm) of poly(*p*-phenylene).<sup>1</sup>

The protonation of the polymers seems to cause the twisting of the main chain, leading to the shift to a lower wavelength (Figure 6).<sup>12</sup> **Polymer-1** and **Polymer-3a** were photoluminescent. **Polymer-1** showed emission peaks at 525 and 552 nm with a quantum yield of 7% in  $\text{HCOOH}$ , and **Polymer-3a** gave rise to an emission peak at 518 nm with a quantum yield of 5% in  $\text{HCOOH}$ .

As described above, *i*-PrMgCl is useful for the regioselective metalation of dihalides of thiophene-pyridine compounds and pyrimidine, and new regiocontrolled  $\pi$ -conjugated polymers have been obtained by polycondensation utilizing the formed intermediate. Detailed analysis of functionalities of the obtained polymers is under way.

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## References and Notes

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- 11 If one assumes *a* and *b* dimensions of 7.4 and 6.9 Å, respectively, for the herringbone packing, *d* spaces of (110), (200), and (210) diffractions are calculated as 5.05, 3.7, and 3.26 Å, respectively, which agree with the observed diffraction peaks. If **Polymer-2** has a repeating height of 4.2 Å along the polymer chain, similarly to poly(*p*-phenylene),<sup>1</sup> the density of **Polymer-2** is calculated as  $d_{\text{calcd}} = 1.21 \text{ g cm}^{-3}$ , which roughly agrees with the observed density  $d_{\text{obs}} = 1.15 \text{ g cm}^{-3}$ . Polymers often give a lower observed density than the calculated density because they contain amorphous parts. The postulated *a* and *b* dimensions or the cross section ( $S = 51 \text{ \AA}^2$ ) of the pyrimidine unit is larger than those of poly(*p*-phenylene) (**PPP**) ( $a = 7.8 \text{ \AA}$  and  $b = 5.5 \text{ \AA}$ ;  $S = 43 \text{ \AA}^2$ )<sup>1</sup> and poly(2,2'-bipyridine-5,5'-diyl) (**PBpy**) ( $a = 8.1 \text{ \AA}$  and  $b = 5.7 \text{ \AA}$ ;  $S = 46 \text{ \AA}^2$ ),<sup>6</sup> thus giving the order:  $S(\text{Polymer-2}) > S(\text{PBpy}) > S(\text{PPP})$ . The importance of the presence of a lone pair of electrons for the determination of the effective cross section of the aromatic unit is suggested.
- 12 The repulsion between a lone pair of electrons<sup>11</sup> and *o*-H may also cause the shift to a shorter wavelength.